## Initial Stages of Water Adsorption on Au Surfaces

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## Received July 11, 1997

We report the results of low-temperature ultrahigh vacuum (UHV) scanning tunneling microscopy (STM) measurements examining the initial stages of water adsorption on Au surfaces. Understanding the adsorption of H<sub>2</sub>O molecules at electrode surfaces in electrolyte solution is one of the most important issues in elucidating the nature of solid-liquid interfaces. Adsorption of H<sub>2</sub>O on metal surfaces has been intensively investigated by means of low-temperature UHV-based experiments.<sup>1,2</sup> Temperature-programmed desorption spectroscopy (TPD),<sup>3-5</sup> work function measurements,<sup>6</sup> and ultraviolet spectroscopy and X-ray photoelectron spectroscopy (UPS and XPS)<sup>7,8</sup> have combined to provide a macroscopic picture of water adsorption on many metal surfaces. There is considerable evidence<sup>1</sup> that H<sub>2</sub>O orders on the (111) planes of Pt and other hydrophilic materials, forming a hexagonal monolayer structure (known as the "bilayer") essentially equivalent to the basal plane of ice (Ih). For example, recent STM images showed wellordered H<sub>2</sub>O films formed on Pt(111) at 140 K.9,10 However, there is less understanding about the initial stages of H<sub>2</sub>O film formation on more weakly interacting surfaces such as Au. This lack of understanding is particularly unfortunate given the variety of surface electrochemical studies utilizing Au(111) surfaces.

Before exposure to H<sub>2</sub>O, STM images<sup>11</sup> of a clean, wellordered Au(111) surface<sup>12</sup> obtained at 100 K show the characteristic "herringbone" pattern with a corrugation amplitude of 0.02 nm arising from the well-studied ( $\sqrt{3} \times 23$ ) Au(111) surface reconstruction.<sup>13</sup> Higher resolution images reveal the underlying Au(111) surface structure.

Figure 1A shows a STM image of the reconstructed Au(111)

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(12) Au(111) surfaces were obtained as Au films deposited on borosilicate glass (Dirk Schröer, Inc., Berlin). The samples were prepared in UHV by cycles of ion sputtering (1 keV,  $P_{N2} = 2 \times 10^{-8}$  mbar) and annealing at temperatures where the sample started to glow. This procedure produced a clean and atomically flat Au(111) surface as checked by AES and STM. Only one heating—sputtering cycle was necessary for routine cleaning. The samples were cooled to 100 K by a cryostat filled with liquid helium.

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**Figure 1.** A. The 202 nm × 202 nm STM image of Au(111) at 100 K following dosing with  $1.5 \times 10^{15}$  molecules/cm<sup>2</sup> of H<sub>2</sub>O.  $V_{tip} = -2.1$  V,  $I_{tunnel} = 0.3$  nA. B. The 78 nm × 77 nm STM image of the same film at 100 K.  $V_{tip} = -2.1$  V,  $I_{tunnel} = 0.1$  nA. C. Cross-sectional view obtained along dark line shown in Figure 1B showing height of the H<sub>2</sub>O islands and clusters.

surface at 100 K after a H<sub>2</sub>O exposure<sup>14</sup> of  $1.5 \times 10^{15}$  molecules/ cm<sup>2</sup>. Clearly evident in the image are many small islands associated with H<sub>2</sub>O exposure, which form on the terraces as well as at step edges. Some of islands are isolated, while others are connected to form larger islands. These islands, surprisingly, evince planar rather than three-dimensional or spherical contours. Although the islands do not exhibit highly dendritic shapes, the island ledges are atomically rough rather than 3-fold symmetric as would be expected for an epitaxial film.

Figure 1B,C shows a close-up STM image taken on an another area of the same surface and a corresponding cross-sectional view. The image shows regions of bare reconstructed Au(111), regions of the planar film, and regions where clusters of a few nanometers in diameter are growing atop the film. No clusters are observed to be growing on the bare Au surface. It is noteworthy that the nucleation of H<sub>2</sub>O islands does not always start from the elbows of the reconstruction as is found for more strongly interacting adsorbates.<sup>15</sup> In this image, small H<sub>2</sub>O

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<sup>(11)</sup> Experiments were performed in an Omicron UHV system (base pressure  $< 2.0 \times 10^{-10}$  mbar) equipped with Auger electron spectroscopy (AES), low-energy electron diffraction (LEED), and an Omicron variable temperature STM. The STM tip was prepared by electrochemical etching of a tungsten wire at 15–20 V ac in a 1 M KOH solution. The bias voltage is that of the tip referred to the sample. The tips were further treated in UHV by applying a high bias voltage (10 V) for several scan lines while scanning on sample areas different from those studied. STM images were recorded at 100 K in height mode.

<sup>(14)</sup> H<sub>2</sub>O adsorption was performed by filling the chamber with a H<sub>2</sub>O pressure of  $1 \times 10^{-9}$  to  $5 \times 10^{-9}$  mbar while maintaining the sample temperature at 100 K. The STM tip was removed from the sample during water exposure to avoid shadowing effects.

islands are seen sitting on the paired reconstruction rows. The row directions of the reconstruction, although hidden as they pass under the islands, do not appear to be altered by the presence of the H<sub>2</sub>O layer. This observation implies that the Au(111) surface remains reconstructed and that, consequently, the strength of the interaction between H<sub>2</sub>O and Au is weak.

The cross-sectional view (Figure 1C) clearly shows that the H<sub>2</sub>O islands do not have atomically flat surfaces. The height of the planar film is about 0.15 nm above the Au surface, while the cluster reaches to 0.45 nm. The measured 0.15 nm thickness of the film agrees well with that expected for the thickness of a monolayer H<sub>2</sub>O film for the tunneling conditions used (ca. 0.1-0.15 nm depending on orientation).<sup>10</sup> Since we do not observe any well-ordered atomic structures on the H<sub>2</sub>O islands-but can obtain atomic resolution from the bare Au surface-these images suggest that the first layer of the H<sub>2</sub>O film forms an amorphous phase. Similar results were obtained on an Au(100) surface.16

The amorphous first layer structure for H<sub>2</sub>O observed here on Au(111) (and Au(100))<sup>16</sup> is consistent with information obtained from LEED measurements on surfaces other than Au, for which this structural information is not available. TPD curves from  $H_2O$  on  $Au(111)^3$  are very similar to those obtained for H<sub>2</sub>O on  $Ag^{17-20}$  and  $Cu^{21,22}$  surfaces in that they exhibit only one desorption peak. In agreement with our STM results, the results from LEED studies for H<sub>2</sub>O on Ag<sup>17-20</sup> and Cu<sup>21,23</sup> suggest that thin films grown on these substrates are also amorphous. In contrast, at least two peaks are observed in TPD from Pt(111) and Rh(001) which have stronger H<sub>2</sub>O-metal interactions.1 These substrates evince ordered H<sub>2</sub>O monolayers.9,24 On Pt(111), H2O forms quasi-one-dimensional chains of molecules at the upper side of step edges;<sup>9</sup> these were never seen on Au(111) or Au(100)—another indication that the  $H_2O$ — Au interaction is much weaker than that found for Pt.

The STM data provide considerable insight into the likely molecular mechanism of growth of H<sub>2</sub>O on Au surfaces at low temperatures. Of interest is the appearance of H<sub>2</sub>O clusters only atop the amorphous film and not on the bare Au(111) surface. This observation suggests that the amorphous film is a necessary precursor to H<sub>2</sub>O multilayer formation. The growth process can be depicted as involving at least two steps: (1) formation of planar, amorphous H<sub>2</sub>O films followed by (2) attachment of three-dimensional H2O clusters atop this film. The STM

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observations indicate that the planar film formed in the first step does not thicken beyond a monolayer. Although the TPD data for H<sub>2</sub>O on Au(111) show a single desorption peak,<sup>3</sup> our results suggest that the H<sub>2</sub>O molecules are adsorbed on Au in at least two states at submonolayer coverages. Thus, desorptionfocussed TPD results may not adequately describe the growth process. The STM results presented here further suggest that individual H<sub>2</sub>O admolecules are highly mobile on bare Au(111) and Au(100) while the mobility of admolecules is suppressed not only at the island ledges but also on top of the planar film.

Our model is anticipated by a vibrational spectroscopic study<sup>25</sup> of  $H_2O$  on Cu(111) which suggested the formation of metastable water species ("wetting species"<sup>26</sup>) which at least partially wets the surface. In accord with this inference, the planar H<sub>2</sub>O films obtained in our observations show direct evidence of a "wetting species". A similar proposal has been made for H<sub>2</sub>O films grown on organic monolayers.<sup>27,28</sup> We note that annealing H<sub>2</sub>O films grown on these and other surfaces at between 120 and 160 K leads to the formation of crystalline ice;<sup>29</sup> annealing was not attempted in the present study. However, our study disagrees with recent interpretations of TPD data<sup>5</sup> which suggested that the growth of H<sub>2</sub>O on Au(111) occurs only through the formation of three-dimensional nanoclusters: while clusters may form following annealing at ca. 140 K, the observation of clusters only atop the planar film indicates the importance of the film in the growth of multilayer H<sub>2</sub>O in the low-temperature regime.

Using a low-temperature STM, we have observed the initial stages of H<sub>2</sub>O adsorption on a Au(111) surface. We show that H<sub>2</sub>O first adsorbs on Au as a planar, amorphous, monolayerhigh film. We do not observe H<sub>2</sub>O cluster formation on the bare metal. Rather, the subsequent growth of H<sub>2</sub>O clusters occurs atop this film. This study shows that, even for a weakly interacting surface such as Au, a 'wetting species' is a necessary precursor to the formation of H<sub>2</sub>O multilayers. Examination of the way adsorbates and annealing alter the H<sub>2</sub>O structure is work in progress.

Acknowledgment. The assistance of Dr. Vania Petrova and the staff of the Center for the Microanalysis of Materials is gratefully acknowledged. This work was funded by Department of Energy grant DE-FG02-91ER45349 through the Materials Research Laboratory at the University of Illinois. STM images were obtained in the Center for the Microanalysis of Materials at the University of Illinois which is supported by Department of Energy grant DE-FG02-91ER45349.

## JA972322H

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